

## C-BN Single-Walled Nanotubes from Hybrid Connection of BN/C Nanoribbons: Prediction by *ab initio* Density Functional Calculations

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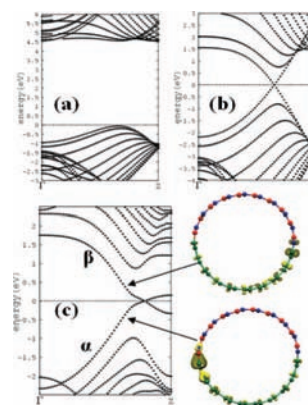
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Subsequent to the discovery of carbon nanotubes (CNTs),<sup>1</sup> ternary boron carbonitride nanotubes (BCN-NTs) attracted increasing interest due to their unique electronic properties and potential applications in nanotechnology.<sup>2</sup> A prime advantage is that the band structure of BCN-NTs can be controlled by changing their atomic composition and configuration.<sup>3,4</sup> Experimentally, B and/or N doping and direct synthesis methods have been developed for synthesizing such novel BCN-NTs.<sup>2,5</sup> Hitherto, most studies of BCN-NTs have focused on random or ordered doping into single-walled carbon nanotubes (SWCNTs), synthesis of sequential BN-C nanodomains along the tube axis, or BN-C coaxial multiwalled NTs.<sup>2,4–6</sup>

Recently, low-dimensional graphene nanoribbons (GNRs) have been fabricated experimentally.<sup>7</sup> These developments have opened up exciting opportunities for the design of novel electronic devices and interconnects.<sup>8</sup> On the other hand, single hexagonal BN nanoribbons (BNNRs) have been also studied extensively.<sup>9</sup> C and BN single-walled nanotubes (SWNTs) can be considered to be rolled from GNRs and BNNRs, respectively.<sup>10</sup> An intriguing question, however, is whether it is possible to form novel C-BN NTs via hybrid connection of a BNNR with a GNR? To explore this question, we report below a series of calculations to study the  $C_x(BN)_{1-x}$  NTs constructed from a GNR/BNNR hybrid connection. We find that armchair and zigzag  $C_{0.5}(BN)_{0.5}$  SWNTs are gapless and narrow gap semiconductors, respectively. Most interestingly, *ab initio* molecular dynamics (AIMD) simulations reveal that a  $C_{0.5}(BN)_{0.5}$  armchair SWNT can be spontaneously formed via the interaction of a BNNR and a GNR at room temperature. Our results suggest an interesting route for synthesizing specific types of BNC-NTs, which may have potential application in BNC-based nanodevices.

A series of single-walled  $C_x(BN)_{1-x}$  (10, 10) armchair NTs with a varied ratio of GNR/BNNR ( $x = 0.1–1.0$ ) was built, and all the geometries were first optimized before calculating their bandstructures<sup>11</sup> (see computational details in the Supporting Information, SI). The  $C_x(BN)_{1-x}$  SWNTs remained largely undeformed due to a small lattice mismatch between BN and graphene. The C–B and C–N bond lengths at the heterointerface are calculated to be 1.531 and 1.387 Å, respectively. Remarkably, when  $x$  is greater than 0.3 and less than 0.9,  $C_x(BN)_{1-x}$  (10, 10) nanotubes displayed negligible band gaps around the Fermi level, similar to the behavior observed



**Figure 1.** Band structure for (a) BN, (b) C, and (c)  $C_{0.5}(BN)_{0.5}$  (10, 10) nanotube and band decomposed charge density for highest occupied valence band ( $\alpha$ ) and lowest unoccupied conduction band ( $\beta$ ). Red, blue balls and yellow surface represent B, N atoms and iso-surface, respectively.

in graphene. The band gap is open if  $x$  is smaller than 0.3, which is in good agreement with a recent study by Wang et al. reporting the conversion of metallicity in SWCNTs into semiconductivity by B/N Codoping.<sup>5</sup>

Figure 1a–c presents band structures for (10, 10) BN, C, and  $C_{0.5}(BN)_{0.5}$  armchair NTs, respectively. Clearly, the BN-NT is a wide gap semiconductor while the armchair CNT shows metallic behavior. However, two linear bands in the CNT became flat (localized) around the Fermi level in the  $C_{0.5}(BN)_{0.5}$  nanotube. It can therefore be inferred that the conductance will be greatly enhanced. To explore the origin of this phenomenon, the partial charge density for the highest occupied band ( $\alpha$ ) and lowest unoccupied band ( $\beta$ ) were plotted as shown Figure 1c. Clearly, the C–B interface affects the valence band maximum ( $\alpha$ ), and a flat minimum conduction band ( $\beta$ ) is attributed to the C–N interface. We have also computed the band structure for (6, 6) and (16, 16)  $C_{0.5}(BN)_{0.5}$  SWNTs with very similar results (see SI); hence it is expected that the observed phenomenon of gapless bands is essentially independent of the diameter of the armchair  $C_{0.5}(BN)_{0.5}$  SWNTs.

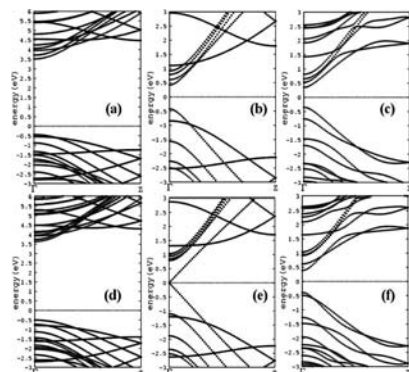
We now focus on (10, 0) and (12, 0)  $C_x(BN)_{1-x}$  zigzag SWNTs with different hybrid ratios of GNR/BNNR. Figure 2 presents the calculated band structures for BN, C, and  $C_{0.5}(BN)_{0.5}$  (10, 0) and (12, 0), respectively. Clearly, there is no major difference between the electronic structures of the (10, 0) and (12, 0) BN-NTs, whereas (10, 0) and (12, 0) C-NTs show a striking contrast (semiconducting and metallic). However, the band structures for (10, 0) and (12, 0)  $C_{0.5}(BN)_{0.5}$  SWNTs are quite similar. The calculated band gaps are

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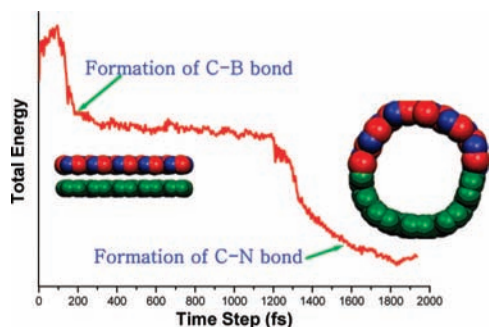
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**Figure 2.** Band structures for a (10, 0) (top panel) and a (12, 0) (bottom panel) BN, C, and  $C_{0.5}(BN)_{0.5}$  zigzag nanotube.



**Figure 3.** Variation of total energy as a function of Car–Parrinello molecular dynamics time step.

0.703 and 0.774 eV, respectively. On the basis of these preliminary results, it may therefore be anticipated that all the zigzag  $C_{0.5}(BN)_{0.5}$  SWNTs should possess a narrow direct band gap (less than 1.0 eV). We have also computed band gaps for  $C_x(BN)_{1-x}$  zigzag SWNTs at different  $x$  ( $x = 0.1–1.0$ ) and found these to range from 0.4 to 0.8 eV (see SI). These consistently small band gaps indicate that  $C_x(BN)_{1-x}$  zigzag SWNTs may be promising as semiconductors for visible light emission as in the case of a layered  $BC_2N$  compound.<sup>12</sup>

Having studied the electronic structure of these novel  $C_x(BN)_{1-x}$  NTs, it is also relevant to explore a possible synthetic route using AIMD simulations. To this end, a bare 10-GNR and bare 10-BNNR (both preoptimized) with zigzag edges at the open ends are positioned at a large separation of 4 Å, such that the interaction between them is initially very weak. AIMD (Car–Parrinello) simulation is then performed at room temperature.<sup>13</sup> Figure 3 presents the variation of the total energy as a function of the AIMD time step. The C–B and C–N bonds are formed at ~200 and 1400 fs, respectively, and  $C_{0.5}(BN)_{0.5}$  in a tubular form evolves at ~2 ps. This indicates that the reaction barrier should be very small if there is any at all; i.e., the formation of armchair  $C_{0.5}(BN)_{0.5}$  SWNTs via the hybrid GNR/BNNR connection is both thermodynamically and kinetically favorable. In contrast, we could not observe the formation of a (10,0)  $C_{0.5}(BN)_{0.5}$  zigzag SWNT from the hybrid connection of the corresponding armchair-type BNNR and GNR partners within 2 ps. Limitations of computational feasibility prevent us at this point from pushing to longer AIMD run times; however, the implication is that the activation barrier should be significant at room temperature. This contrasting reactivity can be understood by the fact that a flat band across the Fermi level occurs in the

case of graphene nanoribbons with zigzag edges, which will induce higher chemical reactivity in this case as found previously.<sup>10</sup>

In summary, we have demonstrated by ab initio density functional calculations that  $C_{0.5}(BN)_{0.5}$  armchair SWNTs are gapless semiconductors, independent of tube diameter. Most interestingly, the direct synthesis (in vacuum) of armchair  $C_{0.5}(BN)_{0.5}$  via the hybrid connection of GNR/BNNR partners is found to be both thermodynamically and dynamically favorable. Armchair  $C_{0.5}(BN)_{0.5}$  NTs can be spontaneously formed via the hybrid connection of a BNNR and a GNR at room temperature. Such novel armchair  $C_{0.5}(BN)_{0.5}$  SWNTs should possess enhanced conductance, which is comparable to graphene. Additionally, the zigzag  $C_{0.5}(BN)_{0.5}$  SWNTs are narrow direct band gap semiconductors (<1 eV), which may potentially be used as new light emitting materials. In light of recent experimental progress and the enhanced degree of control in the synthesis of GNRs and BNNRs,<sup>8,9</sup> our results suggest an interesting avenue for exploration to synthesize novel types of C-BN nanotubes with tunable electronic properties, leading to potentially important applications in BNC-based nanodevices.

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**Supporting Information Available:** Computational details, band structures for (10, 10) and (10, 0)  $C_x(BN)_{1-x}$  SWNTs at different  $x$ , (6, 6) and (16, 16)  $C_{0.5}(BN)_{0.5}$  SWNTs, a movie depicting formation of  $C_{0.5}(BN)_{0.5}$  armchair SWNT via hybrid connection of BN/C nanoribbons during AIMD simulation. This material is available free of charge via the Internet at <http://pubs.acs.org>.

## References

- (1) Iijima, S. *Nature* **1991**, *354*, 56.
- (2) (a) Kawaguchi, M. *Adv. Mater.* **1997**, *9*, 8. (b) Stephan, O.; Ajayan, P. M.; Colliex, C.; Redlich, P.; Lambert, J. M.; Bernier, P.; Lefin, P. *Science* **1994**, *266*, 1683.
- (3) (a) Kim, S. Y.; Park, J. H.; Choi, H. C.; ahn, J. P.; Hou, J. Q.; Kang, H. S. *J. Am. Chem. Soc.* **2007**, *129*, 1705. (b) Miyamoto, Y.; Rubio, A.; Cohen, M. L.; Louie, S. G. *Phys. Rev. B* **1994**, *50*, 4976.
- (4) (a) Blase, X.; Charlier, J.-Ch.; De Vita, A.; Car, R. *Appl. Phys. Lett.* **1997**, *70*, 197. (b) Blase, X.; Charlier, J.-Ch.; De Vita, A.; Car, R. *Appl. Phys. A* **1999**, *68*, 293. (c) Azevedo, S.; de Paiva, R.; Kaschny, J. R. *J. Phys.: Condens. Matter* **2006**, *18*, 10871. (d) Azevedo, S.; de Paiva, R. *Europhys. Lett.* **2006**, *75*, 126.
- (5) Wang, W. L.; Bai, X. D.; Liu, K. H.; Xu, Z.; Golberg, D.; Bando, Y.; Wang, E. G. *J. Am. Chem. Soc.* **2006**, *128*, 6530. (b) Xu, Z.; Lu, W. G.; Wang, W. L.; Gu, C. Z.; Liu, K. H.; Bai, X. D.; Wang, E. G.; Dai, H. J. *Adv. Mater.* **2008**, *20*, 3615.
- (6) (a) Enouz, S.; Stephan, O.; Cochon, J. L.; Colliex, C.; Loiseau, A. *Nano Lett.* **2007**, *7*, 1856. (b) Raidongia, K.; Jagadeesan, D.; Upadhyay-Kahaly, M.; Waghmare, U. V.; Pati, S. K.; Eswaramoorthy, M.; Rao, C. N. R. *J. Mater. Chem.* **2008**, *18*, 83.
- (7) (a) Novoselov, K. S.; Geim, A. K.; Morozov, S. V.; Jiang, D.; Zhang, Y.; Dubonos, S. V.; Grigorieva, I. V.; Firsob, A. A. *Science* **2004**, *306*, 666. (b) Berger, C.; Song, Z. M.; Li, X. B.; Wu, X. S.; Brown, N.; Naud, C.; Mayo, D.; Li, T. B.; Hass, J.; Marchenkov, A. N.; Conrad, E. H.; First, P. N.; de Heer, W. A. *Science* **2006**, *312*, 1191.
- (8) (a) Falko, V. *Nat. Phys.* **2007**, *3*, 151. (b) Geim, A. K.; Novoselov, K. S. *Nat. Mater.* **2007**, *6*, 183.
- (9) (a) Barone, V.; Peralta, J. E. *Nano Lett.* **2008**, *8*, 2210. (b) Park, C. H.; Louie, S. G. *Nano Lett.* **2008**, *8*, 2200. (c) Du, A. J.; Smith, S. C.; Lu, G. Q. *Chem. Phys. Lett.* **2007**, *447*, 181. (d) Chen, Z. G.; Zou, J.; Liu, G.; Li, F.; Wang, Y.; Wang, L. Z.; Yuan, X. L.; Sekiguchi, T.; Cheng, H. M.; Cheng, Lu, G. Q. *ACS Nano* **2008**, *2*, 2183.
- (10) Du, A. J.; Smith, S. C.; Lu, G. Q. *Nano Lett.* **2007**, *7*, 3349.
- (11) (a) Kresse, G.; Furthmüller, J. *Comput. Mater. Sci.* **1996**, *6*, 15. (b) Kresse, G.; Furthmüller, J. *Phys. Rev. B* **1996**, *54*, 11169.
- (12) (a) Watanabe, M. O.; Itoh, S.; Sasaki, T.; Mizushima, K. *Phys. Rev. Lett.* **1996**, *77*, 187. (b) Note that the real band gap is expected to be larger than that predicted in a LDA-DFT calculation since LDA is well known to underestimate band gaps.
- (13) (a) CPMD, Copyright IBM Corp 1990–2006, Copyright MPI für Festkörperforschung Stuttgart 1997. (b) Car, R.; Parrinello, M. *Phys. Rev. Lett.* **1985**, *55*, 2471.

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